

Formation of oriented polypeptides on Au(111) surface depends on the secondary structure controlled by peptide length

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Abstract: We synthesized three different lengths of poly(L-lysine) containing an -SH group at the terminal (PLL_n-SH, *n* (polymerization degree) = 4, 10, 30) and adsorbed them on an Au(111) surface. To analyze the formation process and the structure of self-assembled monolayers (SAMs), we used atomic force microscopy (AFM) and Fourier transform infrared reflection absorption spectra (FT-IR RAS). At the initial stage of SAM growth, formation of nanosize domains was confirmed by AFM imaging. The α -helical PLL₃₀-SH exhibited a well-defined SAM structure after adsorption reached equilibrium. The α -helical PLL₃₀-SH was almost perpendicular to the gold surface and exhibited interesting molecular packing due to the secondary structure of PLL₃₀-SH and the underlying Au(111) array. The tilt angle of the helix axis from the substrate normal was estimated to be about 50° (AFM) and 44° (FT-IR RAS) respectively. On the other hand, PLL₄-SH and PLL₁₀-SH formed β -sheet-type SAMs on the Au(111) surface based on the structure determined by FT-IR RAS spectrum. Copyright © 2005 European Peptide Society and John Wiley & Sons, Ltd.

Keywords: poly(L-lysine); self-assembled monolayer (SAM); atomic force microscopy (AFM); Fourier transform infrared reflection absorption spectra (FT-IR RAS); α -helical structure; β -sheet structure

INTRODUCTION

For two-dimensional integration of molecules, a wide range of uses as functional interface has been considered including applications such as sensing and surface property modification [1–3]. Therefore, there is a need for precision control over the chemical structure of molecules to contribute to the functionality of the two-dimensional integration. In order to construct the structure, the formation of self-assembled monolayer (SAM) with the thiol group has been reported as a convenient method [4–7]. In each such SAM, it has been shown that the intermolecular interaction and the molecules–substrate interactions effect cooperatively and/or interactively to form a highly ordered monolayer structure. This suggests that a novel SAM structure becomes possible by positively incorporating a noncovalent bonding site in a molecular chemical structure, such as a coordinate-bonding and hydrogen-bonding site. However, many problems affect self-organization of molecules into a structure: e.g. the molecular chemical structure, organizational dynamics, the method of evaluation, and immobilization of the SAM.

In this study, we synthesized three different length polypeptides containing a -SH group at the terminal. These polypeptides have many properties such as a secondary structure, optical activity, and

macro-dipole–moment interactions. We believe that using a polypeptide not only provides structural information to the SAMs more precisely but also derives the potential for novel functional interfaces such as aligned thin films and chemical sensors. The synthesized polypeptides were adsorbed on an Au(111) surface in order to characterize their structure and the formation process of SAMs on the Au(111) surface through the use of atomic force microscopy (AFM) and Fourier transform infrared reflection absorption spectrum (FT-IR RAS). We discuss the correlation between the degree of polymerization and the assembled surface structure, and the molecular orientation within the self-assembled molecular layer.

MATERIALS AND METHODS

Solid Phase Peptides Synthesis

Three different lengths of poly(L-lysine) with a -SH moiety at the terminal (PLL_n-SH; Figure 1) were prepared by a peptide synthesizer (P. Biosystems, Pioneer) according to the Fmoc solid phase method [8]. This was carried out on a 0.1-mmol scale using the continuous flow method. Side-chain protection was afforded by trityl for *cys*- and *tert*-butoxycarbonyl for *Lys*. After the final step of deprotection, the resin was dried with 2-propanol. The peptide was cleaved and deprotected by 2.5-h treatment with trifluoroacetic acid (TFA)/water = 95:5 (vol%). It was concentrated *in vacuo* and then added to diethylether. The precipitate was collected by filtration and dried *in vacuo*.

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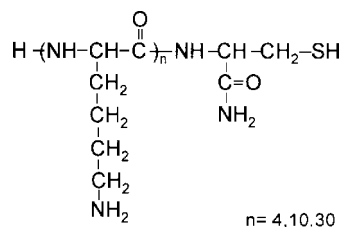


Figure 1 Chemical structure of poly(L-lysine) with thiol group at the terminal.

and a white powder obtained by freeze-drying from a pH 7.0 aqueous solution.

CD Measurement

The circular dichroism (CD) spectra of the polypeptides were measured in trifluoroethanol (TFE) solutions at 20 °C on a JUSCO J-720 CD spectropolarimeter using an optical cell of 1.0 cm path length. The percent of the α -helical contents (f_α) were calculated from Eqn (1) [9].

$$f_\alpha = \frac{([\theta]_{222} + 2000) \times 100}{-30\,000} \quad (1)$$

$[\theta]_{222}$: molar ellipticity of polypeptides at 222 nm.

Preparation of SAM on Au(111) Surface

The Au(111) substrate was prepared by evaporation of gold (99.999%) onto a freshly cleaved mica surface in a homemade vacuum chamber. Evaporation was carried out under previously described conditions where the substrate temperature was kept constant at 300 °C and the pressure in the vacuum chamber was kept at $10^{-6} \sim 10^{-7}$ torr during evaporation [10–12]. After evaporation, the substrate was annealed at 350 °C for 3 h. After cooling the substrate to room temperature, it was treated by flame annealing and immersed directly into TFE solution of PLL_{*n*}-SH to be concentrated to 1.0 mM. The Au(111) substrates were immersed into PLL_{*n*}-SH/TFE solutions for 1 h at 20 °C. The substrate was quickly removed from the solution and immediately rinsed with enough TFE and dried with N₂ gas.

FT-IR RAS Measurement

The FT-IR reflection adsorption spectra of PLL_{*n*}-SH on Au(111) were recorded on a Bio-Rad FTS-6000 Fourier transform infrared spectrometer, and a *p*-polarized beam was obtained through a polarizer. The spectrometer was purged with N₂, and a liquid-nitrogen-cooled mercury-cadmium-telluride (MCT) detector was used for the reflection measurement. The spectra were recorded at a 4 cm⁻¹ resolution with 256 scans. We used a Au(111) surface for background measurement in order to minimize the experimental error.

AFM Measurement

The AFM system used in this study was a NanoScope IIIa (Digital Instruments Inc.). The measurements were performed in the tapping and contact modes (15- μ m scanner) in air at room temperature. For the tapping mode measurement, we

used Si cantilevers (Nanoprobe NCH-10T, Digital Instruments) with a length of 125 μ m. For contact mode measurements, we used a Si₃N₄ cantilever with a spring constant of 0.12 N m⁻¹ (Nanoprobe NP-S, Digital Instruments) for molecular resolution imaging. All the images were collected in the 'height mode', which kept the force constant. In order to minimize drift effect, the system was calibrated for 6 ~ 8 h using a standard mica sample. Images shown in this paper were flattened and plane fitted, but not further manipulated. To determine the thickness of the molecular membrane, the images were analyzed with the 'Fractal Analysis' program supplied by Digital Instruments.

RESULTS AND DISCUSSION

CD Measurements

Figure 2 shows the CD spectra of PLL_{*n*}-SH in TFE at 20 °C. From the CD spectrum of PLL₄-SH, it can be seen that PLL₄-SH mainly forms a random-coil structure in TFE solution at 20 °C. In contrast, the spectra of PLL₁₀-SH and PLL₃₀-SH in TFE show coexisted spectra of the α -helical and random conformations. The α -helix structure has an extremum at 208 nm equal to approximately $-30\,000$ deg cm² dmol⁻¹. For the first approximation, the percent of the α -helix can be calculated from Eqn (1), the α -helical content of PLL₁₀-SH and PLL₃₀-SH were estimated to be about 13 and 90% respectively. Furthermore, the spectrum of PLL₄-SH indicated that PLL₄-SH mainly formed a random-coil structure in the TFE solution. These results indicate that the negative intensity of the molar ellipticity of PLL_{*n*}-SH increased at 222 nm, corresponding to an increase in the degree of polymerization, i. e. the chain length of PLL_{*n*}-SH stabilized the α -helical conformation. Furthermore, in the case of PLL₃₀-SH, the two negative peaks at 208 and 222 nm that are characteristic of an α -helical structure show that the

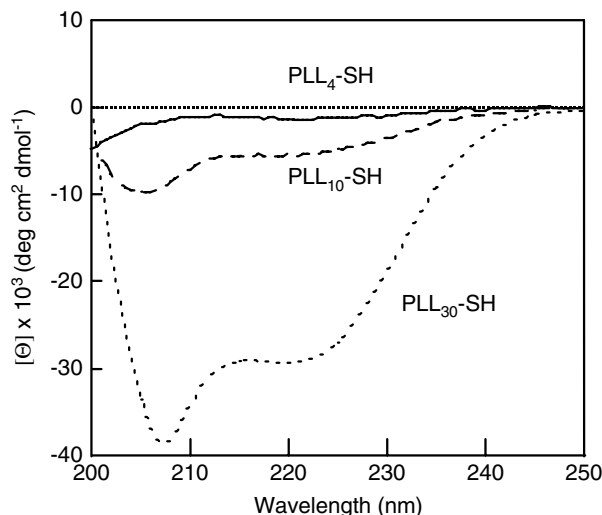


Figure 2 Circular dichroism spectra of PLL_{*n*}-SH in TFE solution at 20 °C.

α -helical polypeptides exist in a dispersed state in TFE solutions [13,14].

FT-IR RAS Measurements

Figures 3–5 show *p*-polarized FT-IR RAS of PLL_{*n*}-SH adsorbed on Au(111) surfaces. From this spectrum of two peaks, we can postulate that the secondary structure of PLL_{*n*}-SH corresponds to the amide I and the amide II bands. Significant differences are observed for the amide bands between the PLL_{4,10}-SH spectrum (Figures 3 and 4) and the PLL₃₀-SH spectrum shown in Figure 5. The amide band positions for PLL₄-SH (amide I = 1687 cm⁻¹, amide II = 1514 cm⁻¹) and PLL₁₀-SH (amide I = 1684 cm⁻¹, amide II = 1521 cm⁻¹) prepared from a 1.0-mM TFE solution for 1 h at room temperature have been attributed to a β -sheet structure on an Au(111) surface [15–17]. On the other hand, the peaks of the PLL₃₀-SH corresponding to the amide I and amide II bands were observed at around 1660 and 1546 cm⁻¹ respectively. This indicated that the peptide was in α -helical conformation on the Au(111) surface and the difference in the intensity ratio between the two peaks (amide I/amide II) can be explained in terms of the orientation of the α -helix peptide on the gold substrate (Scheme 1). The intensity ratio ascribed to the transition moment of amide I and amide II increases as the tilt angle from the surface normal becomes smaller. The relationship between the tilt angle and the intensity ratio of amide I/amide II is given by Eqn (2) [18,19].

$$D_{\text{obs}} = \frac{A_{\text{Iobs}}}{A_{\text{IIobs}}} = K \frac{A_{\text{Ical}}}{A_{\text{IIcal}}} \\ = K \frac{2 \left[\frac{1}{2} (3 \cos^2 \theta - 1) \right] \left[\frac{1}{2} (3 \cos^2 \theta_{\text{I}} - 1) \right] + 1}{2 \left[\frac{1}{2} (3 \cos^2 \theta - 1) \right] \left[\frac{1}{2} (3 \cos^2 \theta_{\text{II}} - 1) \right] + 1} \quad (2)$$

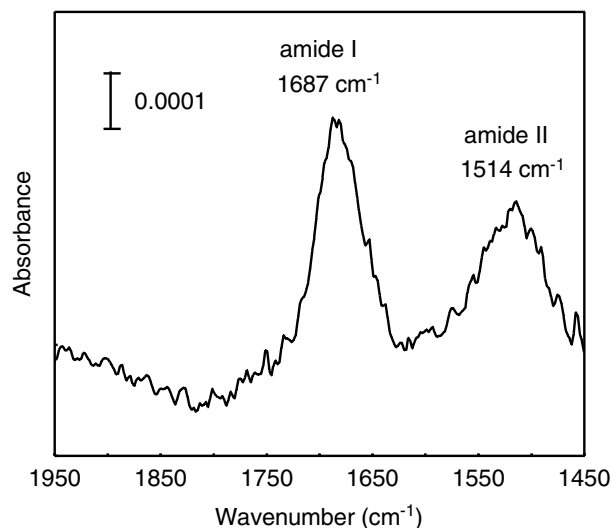


Figure 3 FT-IR RAS spectrum of PLL₄-SH on Au(111) surface.

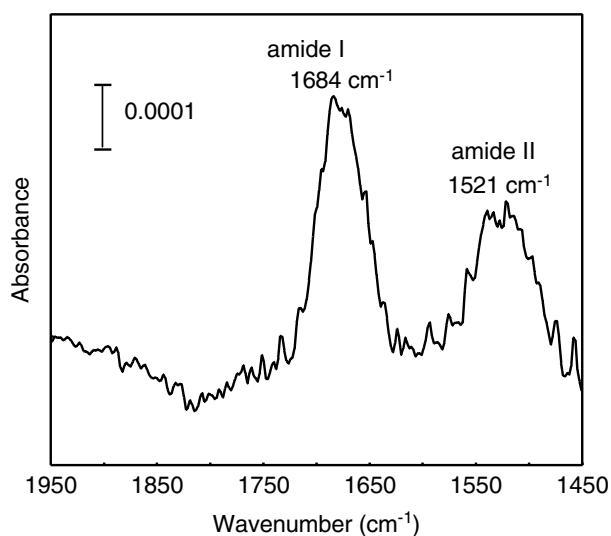


Figure 4 FT-IR RAS spectrum of PLL₁₀-SH on Au(111) surface.

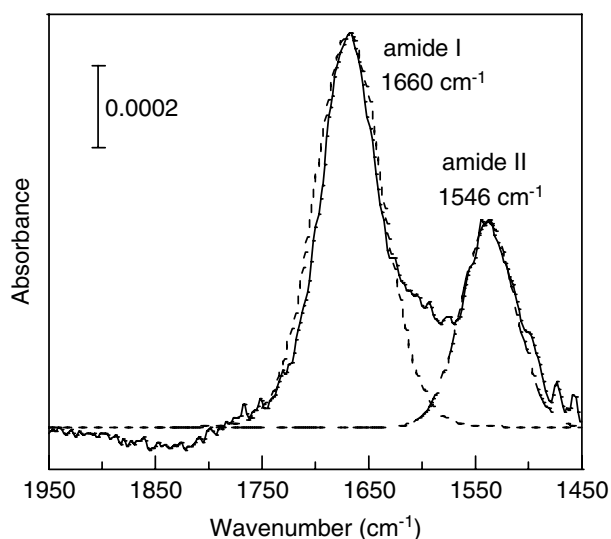


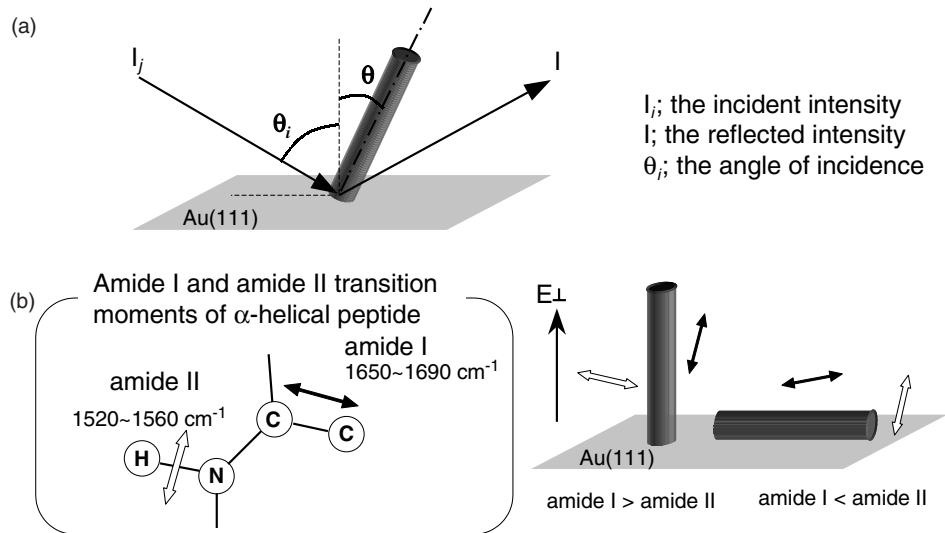
Figure 5 FT-IR RAS spectrum of PLL₃₀-SH on Au(111) surface. The solid line resulted from experimental data. The dotted lines show the amide I and amide II peak area after peak separation.

θ_{I} : the angle between the transition moment of amide I and the axis as 39°.

θ_{II} : the angle between the transition moment of amide II and the axis as 75°.

K : the scaling constant.

To obtain truly quantitative data, Gaussian distribution of the spectrum was used to enhance peak separation and the determination of the peak area. The peak area of the amide I and amide II were estimated to be 0.079 and 0.033 respectively. Hence, intensity ratio (amide I_{obs}/amide II_{obs}) of PLL₃₀-SH SAM was about 2.4. From Eqn (3), the tilt angle was calculated to be about 44° from surface normal. This supports the conclusion



Scheme 1 Schematic illustration of (a) FT-IR RAS measurement and (b) the relationship of amide I and amide II transition moment formed α -helical structure.

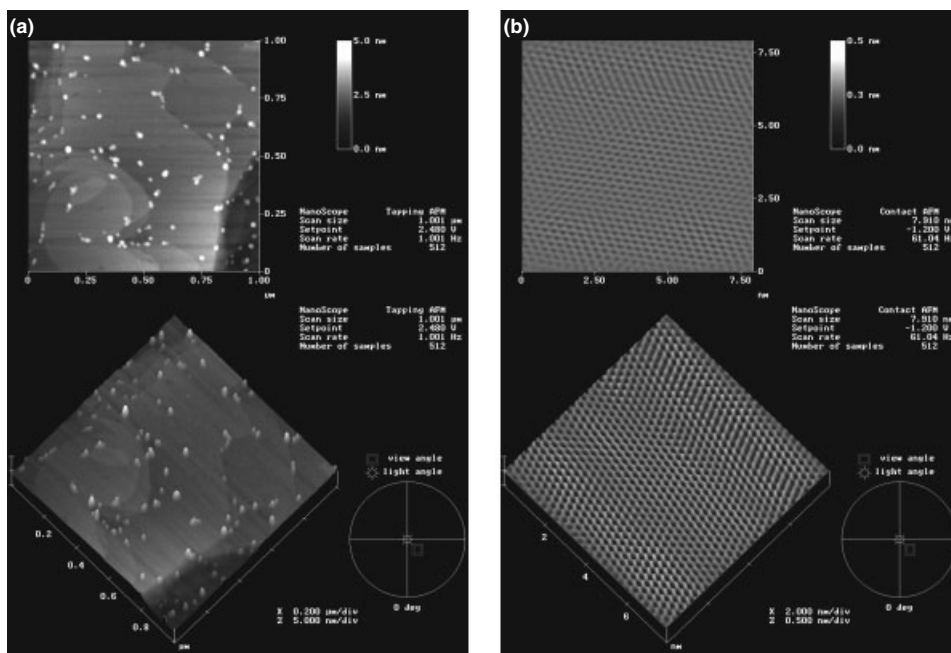


Figure 6 AFM images of Au(111) crystal surface deposited on cleaved mica surface. (a) $1 \mu\text{m} \times 1 \mu\text{m}$, (b) $8 \text{ nm} \times 8 \text{ nm}$. In the images, the bright regions correspond to the high region and the dark regions correspond to the low region.

that the α -helical PLL₃₀-SH was mostly oriented perpendicular to the surface, as described. In addition, the tilt angle of the PLL₃₀-SH helix axis from the substrate normal decreased with the passage of time for adsorption, and a change of the tilt angle was not observed after 1 h.

Surface Morphology of PLL_n-SH on Au(111) Surface

AFM images of the Au(111)/mica substrates used in this study show that the surface morphology of the substrates consisted of flat terraces (Figure 6a). A higher-resolution image of a terrace revealed a

hexagonal pattern of bright spots with nearest neighbor spacing ($3.0 \pm 0.1 \text{ \AA}$), which is consistent with a Au(111) lattice (Figure 6b) [10,11].

The surface morphology of PLL₃₀-SH on Au(111) surface, PLL₃₀-SH formed small domains on the Au(111) surface during the first stage of adsorption. These domains grew to form a homogeneous layer within 1 h, and the Au(111) surfaces were almost covered with molecular layers yielding smooth surfaces (Figure 7c). However, it was also observed that there were several defects in the molecular layers corresponding to

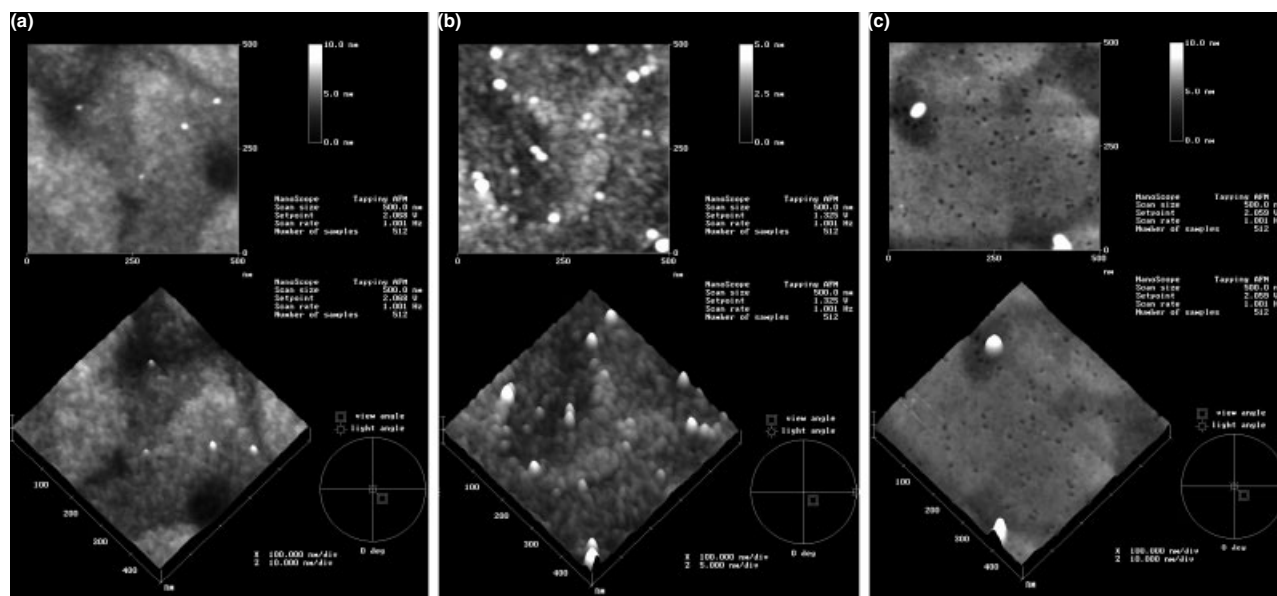


Figure 7 AFM images of PLL_n-SH on Au(111) in 1.0 mM TFE solution for 1 h. (a) PLL₄-SH, (b) PLL₁₀-SH, (c) PLL₃₀-SH.

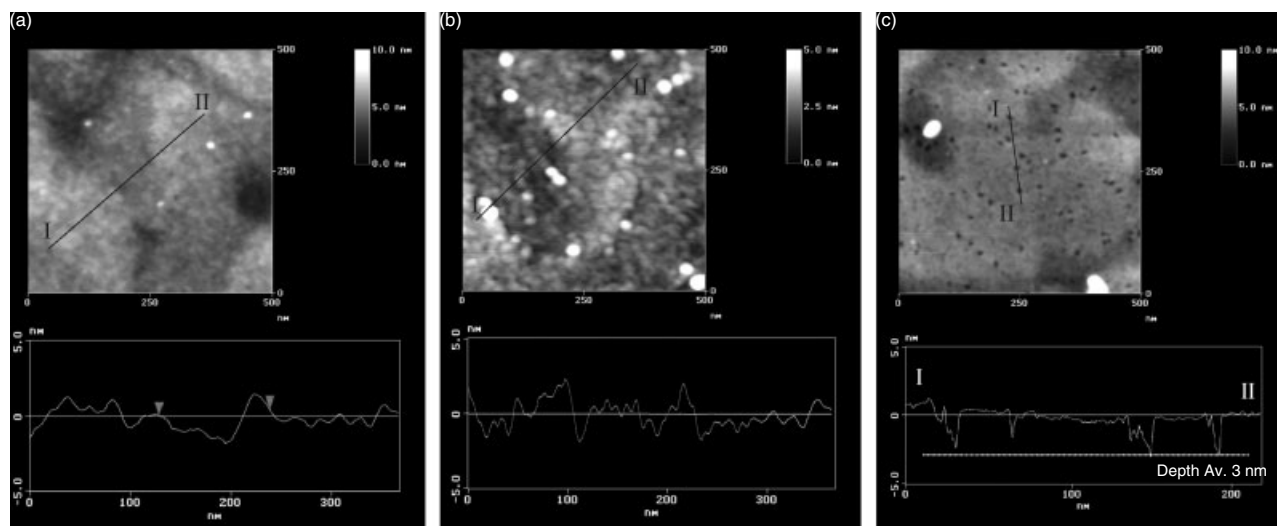


Figure 8 AFM images of PLL_n-SH on Au(111) and cross-sectional profile along the line draw on image. (a) PLL₄-SH, (b) PLL₁₀-SH, (c) PLL₃₀-SH.

the pure Au(111) surface. The difference in height between the smooth surface and the pure Au(111) surfaces provided an estimate of the molecular layer thickness, which was found to be approximately 3.0 nm (Figure 8c). From this distance and the length of PLL₃₀-SH when forming a perfect α -helical structure (4.8 nm) as estimated using the CPK models derived from PEPCON (Figure 9a), we can estimate the tilt angle from the substrate normal, considering the thickness of the monolayer and the length of the α -helical PLL₃₀-SH, a value of 50° (Figure 9b). The molecular orientation of PLL₃₀-SH in SAM must be influenced by substrate–molecule interactions with respect to the underlying Au(111) surface lattice. In this case, the

difference between the diameter of the α -helical PLL₃₀-SH and the distance of the threefold hollow site (the threefold hollow site is the most stable binding site for chemisorptions of a thiol group on Au(111)) causes vacancies among the α -helix rod. Therefore, the α -helical PLL₃₀-SH must form a more closely packed structure at the substrate–air interface to stabilize the vacancies. This causes the tilt of the α -helical rods.

On the other hand, in AFM images of PLL₄-SH and PLL₁₀-SH, we could not observe a homogeneous layer such as a PLL₃₀-SH monolayer (Figure 7a,b). However, from the FT-IR RAS measurements, we concluded that PLL₄-SH and PLL₁₀-SH formed a β -sheet-type SAM on the Au(111) surface. PLL₄-SH and PLL₁₀-SH formed small domains on the Au(111) surface at the first stage

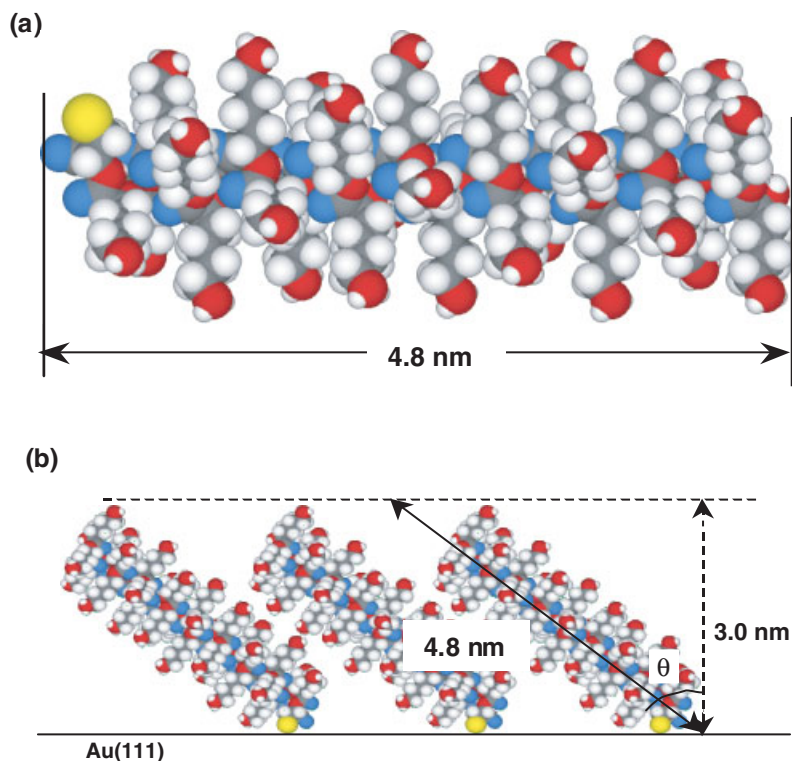


Figure 9 (a) CPK model of α -helical PLL₃₀-SH derived from PEPICON and (b) schematic illustration of α -helical PLL₃₀-SH on Au(111) surface.

of adsorption similar to those seen with PLL₃₀-SH. However, these domains did not grow to a homogeneous layer within 1 h (Figure 8a,b). This could be because of the effects of the secondary structure of polypeptides in TFE solution. PLL₄-SH and PLL₁₀-SH mainly form random-coil structure in TFE solution. Therefore, PLL₄-SH and PLL₁₀-SH did not form rigid molecular structures that would have allowed the construction of a well-defined SAM on the Au(111) surface governed by intermolecular interactions and molecule–substrate interactions. The secondary structure transition to β -sheet structure is probably due to molecular rearrangement. Considering the mobility of the molecules, rearrangement of the molecules seems to have occurred at the physisorption stage before chemisorption resulted in bonding between the thiol group and gold atoms. Because of the intermolecular interactions, PLL₄-SH and PLL₁₀-SH folded into a β -sheet structure at the physisorption stage.

These results indicate that it is possible to construct finely ordered SAMs by making small changes in the peptide, such as the peptide length. Furthermore, the structural features of SAMs constructed from α -helical polypeptides on an Au(111) surface were found to be strongly dependent on factors such as side group interactions and the relationship between the helix structure and underlying Au(111) array. In such a case, we cannot ignore the effect of the solvent used for immersion; however, it was demonstrated that small modifications

to the polypeptide molecules changed the intermolecular interactions that govern the structure of SAMs. This enabled us to construct a novel structure of highly ordered SAMs and to demonstrate their utility as new functional interfaces.

CONCLUSIONS

The structures and growth of SAMs using three different lengths of PLL_{*n*}-SH on Au(111) were studied by AFM and FT-IR RAS measurements. The structures of the PLL_{*n*}-SH SAMs were found to be strongly dependent on the degree of polymerization, i.e. the chain length of the PLL_{*n*}-SH. This resulted in differences in the secondary structures in the TFE solution and in the intermolecular interactions during the physisorption stage.

PLL₄-SH and PLL₁₀-SH formed β -sheet structure on Au(111) surface, and α -helical PLL₃₀-SH assembled into well-defined SAMs on Au(111) surface. The structural features of the SAMs were found to be strongly dependent on several factors, including the relationship between the helix structure and the underlying Au(111) array, the effect of the solvent used, and the side group interaction. This appeared to be because the length of PLL_{*n*}-SH allowed for the rearrangement of the secondary structure at the physisorption stage; therefore, the intermolecular interaction that causes the assembly of the SAM occurred. These results suggest that the secondary structure of the polypeptide

plays a large role in regulating the structure of SAMs through intermolecular interaction.

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